

Fabrication of Single-Walled Carbon Nanotube/Si Heterojunction Solar Cells with High Photovoltaic Performance

Feijiu Wang,[†] Daichi Kozawa,[†] Yuhei Miyauchi,^{†,‡} Kazushi Hiraoka,[§] Shinichiro Mouri,[†] Yutaka Ohno,[⊥] and Kazunari Matsuda^{*,†}

[†]Institute of Advanced Energy, Kyoto University, Uji, Kyoto 611-0011, Japan

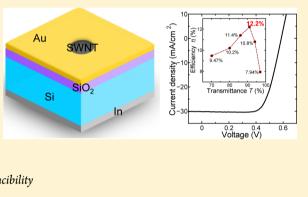
[‡]Japan Science and Technology Agency, PRESTO, 4-1-8 Honcho Kawaguchi, Saitama 332-0012, Japan

[§]Hitachi Zosen Corporation, 2-11, Funamachi 2-Chome, Taisho-ku, Osaka 551-0022, Japan

[⊥]Department of Quantum Engineering, Nagoya University, Chikusa-ku, Nagoya 464-8603, Japan

Supporting Information

ABSTRACT: The photovoltaic properties of carbon nanotube/Si heterojunction solar cells were investigated using network films of high-quality single-walled carbon nanotubes (SWNTs) grown by atmospheric-pressure floating-catalyst chemical vapor deposition. Because of the optimization of the device window size and the utilization of SWNT thin films with both low resistivity and high transparency, a high photovoltaic conversion efficiency of greater than 12% was achieved for SWNT/Si heterojunction solar cells without any postprocessing, such as carrier doping treatment. In addition, the high reproducibility of the photovoltaic performance of these devices in air was demonstrated.



KEYWORDS: solar cell, carbon nanotube, high efficiency, high reproducibility

arbon nanotubes have attracted a great deal of interest for photovoltaic applications because of their excellent optical and electronic properties, including the ability to tune their band gaps over a wide range by controlling the tube diameter¹ and their high carrier mobilities along their one-dimensional axes.²⁻⁴ Solar cells based on simple heterostructures of singlewalled carbon nanotubes (SWNTs) and Si have been extensively studied as model devices.⁵⁻⁹ In the past few years, the photovoltaic performance of SWNT/Si heterojunction solar cells has been improved using postprocessing techniques and complicated device structures, such as carrier doping of SWNT films by the infiltration of acid,¹⁰ controlling the electronic junctions using an ionic liquid electrolyte,^{11,12} and coating the Si surface with an antireflection layer.¹³ Photovoltaic conversion efficiencies of greater than 10% have been reported for these devices. However, the need to use complicated postprocessing techniques is impracticable for the application of SWNTs to large-scale solar cells. In addition, such techniques may lead to less stable devices. Hence, the development of a method for the production of high-efficiency, stable SWNT/Si heterojunction solar cells that do not require postprocessing is strongly desired.

The physical properties of SWNT thin films strongly affect the photovoltaic performance of SWNT/Si heterostructure solar cells. It has been revealed that the SWNT film serves as a semitransparent carrier transport layer in these solar cell devices.¹⁴ Hence, one of the major factors limiting the photovoltaic efficiency of previously reported SWNT solar cells may be related to the carrier transport properties of the films. It is thus expected that SWNT/Si heterostructure solar cells with these films that have high transparency and low sheet resistivity will improve photovoltaic conversion efficiencies.

Here, we report the results of our investigation of the photovoltaic properties of SWNT/Si heterojunction solar cells using SWNT thin films fabricated using atmospheric-pressure floating-catalyst chemical vapor deposition (FC-CVD). A photovoltaic conversion efficiency of ~10% was obtained for solar cells using these high-quality SWNT thin films with both high transparency and low sheet resistivity. In addition, various cells with SWNT films of different active window sizes and thicknesses were examined to determine the effect of these characteristics on the transparency of these films. Because of the optimization of such device parameters and the utilization of a high-quality SWNT film with both low resistivity and high transparency, a high photovoltaic conversion efficiency of greater than 12% was achieved in a SWNT/Si heterojunction solar cell without any additional postprocessing, such as carrier doping. In addition, the high reproducibility of the photovoltaic performance of these devices in air was demonstrated. These are important insights for large-scale carbon nanotube solar cells with high photovoltaic conversion efficiency.

Received: November 21, 2013 Published: March 20, 2014

ACS Photonics

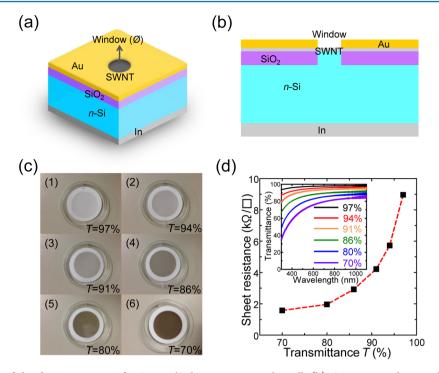


Figure 1. (a) Schematic of the device structure of a SWNT/Si heterojunction solar cell. (b) Cross-sectional view of the device structure. (c) Photographs of SWNT films with transmittance values T = 97, 94, 91, 86, 80, and 70% at 550 nm, respectively. (d) Sheet resistance of SWNT films as a function of *T*. Inset shows the transmission spectra of the SWNT films.

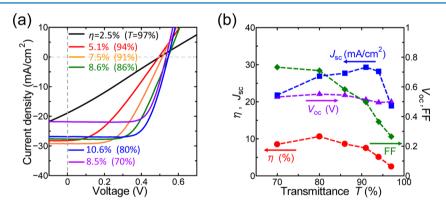


Figure 2. (a) Current density–voltage (J-V) curves for heterojunction solar cells with an active window size of 2 mm fabricated using SWNT films of various thicknesses (transmittance). The conversion efficiency of the solar cells and the corresponding transmittance values for the SWNT films are shown in the figure. (b) Values for η , FF, J_{sc} , and V_{oc} obtained from the J-V curves for each device as a function of T.

RESULTS AND DISCUSSION

Figure 1a shows the schematic of the device structure of a SWNT/Si heterojunction solar cell, and Figure 1b exhibits a cross-section view. Figure 1c shows photographs of uniform and large-area (12.56 cm²) SWNT films fabricated using an atmospheric-pressure FC-CVD technique by feeding a carbon source gas with a catalyst precursor. This technique can control the thickness of SWNT films and is also useful for large-scale SWNT/Si solar cell applications. The SWNTs were collected at times of 12, 16, 20, 30, 40, and 60 min. The corresponding transmittance spectra of these films are shown in the inset of Figure 1d. The transmission value for each film at 550 nm (T)is indicated in the inset of Figure 1d. Figure 1d shows the sheet resistances of the SWNT films as a function of T. The SWNT films fabricated by the FC-CVD method exhibit both low resistivity and high transparency, such as 2 k Ω / for T = 80%, which is significantly superior to those of films prepared using a

spray method.¹⁸ We evaluated the first sub-band transition energy of SWNTs, E_{11} , as 0.65 eV by optical absorption spectrum in Figure S2 and determined the diameter of SWNTs (~1.6 nm) from previous reports.^{16,19}

The effect of the thickness of the SWNT film on the photovoltaic properties of SWNT/Si heterojunction solar cells was investigated. Figure 2a shows the J-V curves of SWNT/Si heterojunction solar cells with an active window size (ϕ) of 2 mm fabricated using SWNT films with different thicknesses (transmission values T). The J-V curves for the SWNT/Si heterojunction solar cells drastically change as the transmittance of the SWNT film varies. On the basis of the evaluation of this series of J-V curves, it is found that the photovoltaic conversion efficiency η strongly depends on the thickness of the film (i.e., the transmittance T). Figure 2b shows the values obtained for η , the fill factor (FF), the short-circuit current density (J_{sc}), and the open-circuit voltage (V_{oc}) as a function of T derived from the series of J-V curves. The

photovoltaic conversion efficiency η increases as *T* decrease and reaches a maximum value at $T \approx 80\%$. In contrast, a negligible change in V_{oc} is observed, while the FF drastically increases as *T* decreases. In addition, J_{sc} drastically increases as *T* decreases in the high-transmittance region and moderately decreases as *T* decreases in the low-transmittance region. Thus, an increase in both FF and J_{sc} from 26.5% to 71% and from 18.9 mA/cm² to 26.9 mA/cm², respectively, contributes to the drastic increase in η from 2.5% to 10.6% from T = 97% to 80%. It should be noted that the photovoltaic conversion efficiency of the cell with a high-transparency and low-resistivity SWNT film prepared using FC-CVD (T = 80%) reached a value of 10.6%, which is much larger than the value of 3.2% obtained for cells prepared using a spray method ($T \sim 80\%$).¹⁸

Next, the change in the conversion efficiency of solar cells as a function of the window size and SWNT film transmittance was investigated. The SWNT/Si heterojunction solar cells with active window sizes (ϕ) ranging from 1 to 3 mm and SWNT films with transmittance (T) ranging from 70% to 97% were prepared (Figures S3–S5). The photovoltaic conversion efficiencies of solar cells with SWNT films of different transmittance are shown in Figure 3 as a function of the

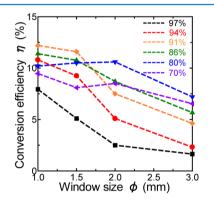


Figure 3. Photovoltaic conversion efficiency of solar cells with SWNT films of various thicknesses as a function of the window size.

window size. The figure shows that the photovoltaic conversion efficiency increases as the active window size decreases, mainly because of a reduction in the effective series resistance, as discussed below. However, the optimized *T* values are different depending on the active window size. Thus, the optimized

device parameters for SWNT/Si heterojunction solar cells ($\phi = 1 \text{ mm}$ and T = 91%) are determined based on these results and are shown to provide a high conversion efficiency of 12.2%.

Here, we discuss the photovoltaic conversion process of these SWNT/Si heterojunction solar cells. A diagram of energy levels of SWNT/Si heterojunction solar cells is shown in Figure S7. According to the previous studies, it was found that the photogenerated electron and hole pairs (carriers) are mainly generated in the n-Si layer.^{6,7} The electrons and holes are separated due to the band alignment at SWNTs and the Si interface, and these electrons and holes as photocarriers contribute to the photovoltaic properties of p-SWNT/n-Si solar cells.

To gain further insight into the impact of the properties of the SWNT film on the photovoltaic performance of SWNT/Si heterojunction solar cells, the series resistance (R_s) of the optimized devices ($\phi = 1 \text{ mm}$) fabricated by different thickness SWNT films was calculated using the *J*–*V* curve data and the following relationship:^{20,23}

$$I\frac{\mathrm{d}V}{\mathrm{d}I} = R_{\mathrm{s}}I\tag{1}$$

Figure 4a shows that the series resistance increases as T increases (decreased thickness of the SWNT network film). This result suggests that the decrease in the FF with an increase in T (Figure 2b) is because of the increase in R_s . The plot of I(dV/dI) for the device fabricated using a SWNT film with T = 91% as a function of I is shown in the inset of Figure 4a. The value for R_s (= 180 Ω) corresponding to 1.41 Ω ·cm² is obtained from the slope of the line in the inset of Figure 4. This low series resistance is mainly because of the high density and close connection of the SWNT network, which occurs because of Y-junction formation in the films fabricated using the FC-CVD method.¹⁶

The diode ideality factor (n) was also evaluated to investigate the carrier transport mechanism of the devices. This factor is determined under forward bias conditions using the relationship described below:²⁰

$$\frac{1}{n} = \frac{kT}{q} \frac{\mathrm{d}\ln I_{\mathrm{dark}}}{\mathrm{d}V} \tag{2}$$

where *q* is the electronic charge, *k* is the Boltzmann constant, *V* is the applied voltage, and I_{dark} is the dark current. The plot of ln I_{dark} for the device fabricated using the film with T = 91% as a

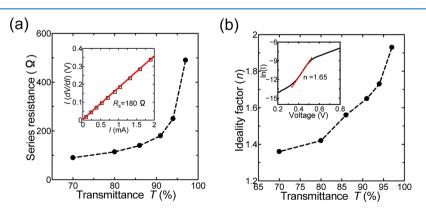


Figure 4. (a) Series resistance of the devices plotted as a function of the transmittance *T*. The inset shows I(dV/dI) as a function of *I* for the device fabricated using the SWNT film with T = 91%. The linear fitted result is shown as a solid line, and the slope of the fitted line corresponds to the cell series resistance. (b) Diode ideality factors as a function of the transmittance *T*. The inset shows the semilogarithmic plot of the *J*–*V* curve for the device fabricated using the SWNT film with T = 91%. The linear fitted line is shown as the solid line.

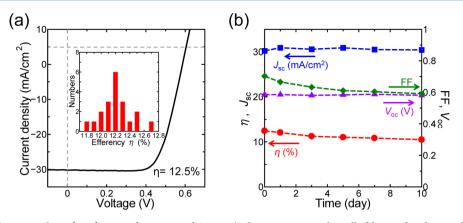


Figure 5. (a) Current density–voltage (*J*–*V*) curves for optimized SWNT/Si heterojunction solar cells fabricated with a window size of 1 mm using a SWNT film with T = 91%. The inset shows a histogram of the variation in the photovoltaic conversion efficiency of the fabricated SWNT/Si solar cell devices. (b) Time evolution of the photovoltaic properties FF, V_{OC} , J_{SC} , and η of an optimally prepared device.

function of *V* is shown in the inset of Figure 4b. From the slope of this line, the diode ideality factor *n* is determined to be 1.65. Figure 4b shows the values for *n* obtained for the devices as a function of *T*. The figure shows that *n* moderately decreases from 1.93 to 1.36 as *T* decreases (i.e., the thickness of the SWNT film increases). These values are much smaller than those previously reported^{18,21} and nearly the same as those of typical Si solar cells.^{22,23} On the basis of these results, it can be concluded that *n* is related to the diffusion recombination process, accompanied by tunneling.²⁴ The moderate increase in *n* with increasing *T* is because of the increase in the sheet resistance of these films.²⁵

Figure 5a shows a typical J-V curve of the optimized SWNT/Si heterojunction solar cell ($\phi = 1 \text{ mm}$) using the SWNT film with T = 91% under AM 1.5 conditions. The values for J_{sc} , V_{oc} and FF extracted from the J-V curve are 30.2 mA/ cm², 0.586 V, and 70%, respectively, which result in a high photovoltaic conversion efficiency η of 12.5%. This value (12.5%) is higher than the previously reported value $(10.5\%)^{11}$ and is the highest photovoltaic conversion efficiency for a SWNT/Si heterojunction solar cell fabricated using as-grown SWNTs without any carrier doping treatment as a postprocessing technique. Previous studies of the photovoltaic conversion process have revealed that the SWNT film mainly serves as a semitransparent carrier transport layer in SWNT/Si heterojunction solar cells.²⁶ Thus, the high photovoltaic conversion efficiency of greater than 12% is because of the high quality (low resistivity and high transparency) of the SWNT film.

The photovoltaic conversion efficiency of 20 solar cells fabricated using the optimized device parameters was then measured, and a remarkable reproducibility for η in the narrow range from 11.8% to 12.7% is observed, as shown in the inset of Figure 5a. The stability of the photovoltaic performance in the SWNT/Si heterojunction solar cell was also examined. Figure 5b shows the time evolution of the photovoltaic conversion efficiency η , J_{sc} , and V_{oc} of the solar cell. The values for J_{sc} and $V_{\rm oc}$ nearly remain the same, while the FF moderately decreases with prolonged time, resulting in a modest decay in the photovoltaic conversion efficiency from 12.5% to 10.5% over 10 days. This decreased rate of 0.2% per day is much smaller than the previously reported value of $\sim 1.2\%$ per day for SWNT/Si solar cells doped with HNO3.15 Thus, the optimized SWNT/Si solar cells fabricated using SWNT films grown by FC-CVD that have low resistivity and high transparency show both superior photovoltaic performance (conversion efficiency >12%) and

high stability. Also, with HNO_3 doping, the optimized SWNT/ Si heterojunction solar cell exhibits a very high photovoltaic conversion efficiency of 14.5% (Figure S8).

CONCLUSION

In summary, SWNT/Si heterojunction solar cells prepared using high-quality SWNT films with low resistivity and high transparency fabricated by the FC-CVD method were investigated. The optimum device structure was determined to consist of a 1 mm diameter window and a SWNT film with T = 91%, and a photovoltaic conversion efficiency of 12% was obtained for a SWNT/Si heterojunction solar cell prepared without postprocessing using this structure. In addition, the high reproducibility of the photovoltaic performance of this type of device was clearly demonstrated, suggesting the superior potential of carbon nanotube solar cells. These are also important steps for large-scale carbon nanotube solar cells with high photovoltaic conversion efficiency.

METHODS

SWNTs were grown using the FC-CVD method, as described previously.^{15,16} Carbon monoxide (CO) was the carbon source, and the catalyst particles were produced by the decomposition of ferrocene vapor. CO (100 sccm) was passed through a temperature-controllable cartridge containing ferrocene powder. Additional CO (300 sccm) and CO₂ (3 sccm) were also introduced to the furnace, and the growth temperature was set at 850 °C. The SWNTs film was collected at room temperature on a membrane filter comprising cellulose acetate and nitrocellulose. The thickness of a SWNT film can be controlled by the collection time.

The n-Si substrates $(1-10 \ \Omega \cdot cm)$ with window sizes $(\phi:$ diameter) ranging from 1 to 3 mm were used for the solar cell devices. The native SiO₂ layer of the active window was removed using a buffered HF solution. The membrane filter containing the SWNT film was then placed on the Si substrate such that the film was in direct contact with the top surface of the substrate. A fiber wiper was then applied to the surface of the substrate with adequate pressure to release the film from the membrane filter and to create a connection between the film and the substrate. It was essential to place a drop of ethanol on the surface to ensure good contact between the films and the substrates and to densify the SWNT film.¹⁷ Next, gold (Au) as the anode was coated around the SWNT film to enclose the active area, while onto the back side of the Si

substrate as the cathode was made. The schematic of the detailed fabrication process of a SWNT/Si solar cell is also illustrated in Figure S1 of the Supporting Information.

The optical transmission spectra of the SWNT films on the glass substrates were measured using a UV/vis/NIR spectrophotometer (Hitachi U-4100). To evaluate the photovoltaic properties of the devices, the SWNT/Si heterojunction solar cells were irradiated using a solar simulator (San-Ei Electric XES-40S1) under AM 1.5 conditions (~100 mW/ cm²), and the current density–voltage (J-V) data were recorded using a sourcemeter (Keithley 2400). The AM 1.5 (100 mW/cm²) conditions in the solar simulator were confirmed using a standard cell (BS-500BK). The sheet resistance of the SWNT network was measured using a digital multimeter (Keithley 2100) connected to a four-probe collinear arrangement (Astellatech SR4-S). To measure the ionization potential for the SWNT film, the photoelectron emission spectrum (RIKEN KEIKI, AC-3) was used.

ASSOCIATED CONTENT

Supporting Information

Additional data on the schematic of the detailed fabrication process of the devices, absorption spectra of SWNT film, photoelectron spectra of SWNT film, schematic of energy band diagram, window size, and HNO_3 doping dependence of photovoltaic properties are shown in the Supporting Information. This material is available free of charge via the Internet at http://pubs.acs.org.

AUTHOR INFORMATION

Corresponding Author

*E-mail: matsuda@iae.kyoto-u.ac.jp.

Notes

The authors declare no competing financial interest.

ACKNOWLEDGMENTS

Y.M. was financially supported by PRESTO, JST. This study was supported by a Grant-in-Aid for Scientific Research from JSPS (Grant Nos. 22740195, 23340085, 24681031, and 40311435), Yazaki Memorial Foundation for Science and Technology, Nippon Sheet Glass Foundation for Materials Science and Engineering, and The Canon Foundation.

REFERENCES

(1) Saito, R.; Dresselhaus, G.; Dresselhaus, M. S. *Physical Properties of Carbon Nanotubes*; Imperial College: London, 1998.

(2) Kanungo, M.; Lu, H.; Malliaras, G. G.; Blanchet, G. B. Suppression of metallic conductivity of single-walled carbon nanotubes by cycloaddition reactions. *Science* **2009**, 323, 234–237.

(3) Fuhrer, M. S.; Kim, B. M.; Durkop, T.; Brintlinger, T. Highmobility nanotube transistor memory. *Nano Lett.* **2002**, *2*, 755–759.

(4) Snow, E. S.; Campbell, P. M.; Ancona, M. G. High-mobility carbon-nanotube thin-film transistors on a polymeric substrate. *Appl. Phys. Lett.* **2005**, *86*, 033105.

(5) Ong, P. L.; Euler, W. B.; Levitsky, I. A. Hybrid solar cells based on single-walled carbon nanotubes/Si heterojunctions. *Nanotechnology* **2010**, *22*, 105203.

(6) Kozawa, D.; Hiraoka, K.; Miyauchi, Y.; Mouri, S.; Matsuda, K. Analysis of the photovoltaic properties of single-walled carbon nanotube/silicon heterojunction solar cells. *Appl. Phys. Express* **2012**, *5*, 042304.

(7) Jung, Y.; Li, X. K.; Rajan, N. K.; Taylor, A.; Reed, M. A. Record high efficiency single-walled carbon nanotube/silicon p-n junction solar cells. *Nano Lett.* **2013**, *13*, 95–99.

(8) Li, Z. R.; Kunets, V. P.; Saini, V.; Xu, Y.; Dervishi, E.; Salamo, G. J.; Biris, A. R.; Biris, A. S. Light-harvesting using high density p-type single wall carbon nanotube/n-type silicon heterojunctions. *ACS Nano* **2009**, *3*, 1407–1414.

(9) Tune, D. D.; Flavel, B. S.; Krupke, R.; Shapter, J. G. Carbon nanotube-silicon solar cells. *Adv. Energy Mater.* 2012, 2, 1043-1055.

(10) Jia, Y.; Cao, A.; Bai, X.; Li, Z.; Zhang, L.; Guo, N.; Wei, J.; Wang, K.; Zhu, H.; Wu, D.; Ajayan, P. M. Achieving high efficiency silicon-carbon nanotube heterojunction solar cells by acid doping. *Nano Lett.* **2011**, *11*, 1901–1905.

(11) Wadhwa, P.; Liu, B.; McCarthy, M. A.; Wu, Z.; Rinzler, A. G. Electronic junction control in a nanotube-semiconductor Schottky junction solar cell. *Nano Lett.* **2010**, *10*, 5001–5005.

(12) Wadhwa, P.; Seol, G.; Petterson, M. K.; Guo, J.; Rinzler, A. G. Electrolyte-induced inversion layer schottky junction solar cells. *Nano Lett.* **2011**, *11*, 2419–2423.

(13) Shi, E.; Zhang, L.; Li, Z.; Li, P.; Shang, Y. Y.; Jia, Y.; Wei, J.; Wang, K.; Zhu, H.; Wu, D.; Zhang, S.; Cao, A. TiO_2 -coated carbon nanotube-silicon solar cells with efficiency of 15%. *Sci. Rep.* **2012**, *2*, 884.

(14) Jeong, S.; Garnett, E. C.; Wang, S.; Yu, Z.; Fan, S.; Brongersma, M. L.; McGehee, M. D.; Cui, Y. Hybrid silicon nanocone-polymer solar cells. *Nano Lett.* **2012**, *12*, 2971–2976.

(15) Moisala, A.; Nasibulin, A. G.; Brown, D. P.; Jiang, H.; Khriachtchev, L.; Kauppinen, E. I. Single-walled carbon nanotube synthesis using ferrocene and iron pentacarbonyl in a laminar flowreactor. *Chem. Eng. Sci.* **2006**, *61*, 4393–4402.

(16) Sun, D.-M.; Timmermans, M. Y.; Tian, Y.; Nasibulin, A. G.; Kauppinen, E. I.; Kishimoto, S.; Mizutani, T.; Ohno, Y. Flexible highperformance carbon nanotube integrated circuits. *Nat. Nanotechnol.* **2011**, *6*, 156–161.

(17) Sun, D.-M.; Timmermans, M. Y.; Kaskela, A.; Nasibulin, A. G.; Kishimoto, S.; Mizutani, T.; Kauppinen, E. I.; Ohno, Y. Mouldable allcarbon integrated circuits. *Nat. Commun.* **2013**, *4*, 2302.

(18) Wang, F.; Kozawa, D.; Miyauchi, Y.; Hiraoka, K.; Mouri, S.; Matsuda, K. Enhancement mechanism of the photovoltaic conversion efficiency of single-walled carbon nanotube/Si solar cells by HNO₃ doping. *Appl. Phys. Express* **2013**, *6*, 102301.

(19) Weisman, R. B.; Bachilo, S. M. Dependence of optical transition energies on structure for single-walled carbon nanotubes in aqueous suspension: an empirical Kataura plot. *Nano Lett.* 2003, *3*, 1235–1238.
(20) Nelson, J. *The Physics of Solar Cells*; Imperial College: London, 2003.

(21) Jia, Y.; Wei, J.; Wang, K.; Cao, A.; Shu, Q.; Gui, X.; Zhu, Y.; Zhuang, D.; Zhang, G.; Ma, B.; Wang, L.; Liu, W.; Wang, Z.; Luo, J.; Wu, D. Nanotube-silicon heterojunction solar cells. *Adv. Mater.* **2008**, 20, 4594–4598.

(22) Green, M. A.; Blakers, A. W.; Shi, J.; Keller, E. M.; Wenham, S. R. 19.1% efficient silicon solar cell. *Appl. Phys. Lett.* **1984**, 44, 1163.

(23) Radziemska, E. Dark I–U–T measurements of single crystalline silicon solar cells. *Energy Convers. Manage.* **2005**, *46*, 1485–1494.

(24) Rakhshani, A. E. Heterojunction properties of electrodeposited CdTe/CdS solar cells. J. Appl. Phys. 2001, 90, 4265.

(25) Xie, C.; Zhang, X.; Wu, Y.; Zhang, X.; Zhang, X.; Wang, Y.; Zhang, W.; Gao, P.; Han, Y.; Jie, J. Surface passivation and band engineering: a way toward high efficiency graphene-planar Si solar cells. *J. Mater. Chem. A* **2013**, *1*, 8567–8574.

(26) Tune, D. D.; Hennrich, F.; Dehm, S.; Klein, M. F. G.; Glaser, K.; Colsmann, A.; Shapter, J. G.; Lemmer, U.; Kappes, M. M.; Krupke, R.; Flavel, B. S. The role of nanotubes in carbon nanotube-silicon solar cells. *Adv. Energy Mater.* **2013**, *3*, 1091–1097.